Coarsening (and anticoarsening) dynamics

Paolo Politi

Institute for Complex Systems National Research Council (CNR), Florence (Italy) Encyclopedia of Complexity and Systems Science (Springer, 2009)

Coarsening -

The monotonic increase of the typical length scale of a structure in time. [...] Coarsening slows down if the length scale increases.

Phase Transitions and Critical Phenomena

Volume 8

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3. The Dynamics of First-order Phase Transitions

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Concentration C

FIG. 1. The coexistence curve (solid line) and classical spinodal curve (dashed line) are shown schematically for a system such as a binary fluid or binary alloy. Typical quenches into the metastable (m) and the unstable (u) regions are also shown. In the former case the system is under-cooled by an amount δT , corresponding to an initial supersaturation $\delta c_1 = c_1 - c_A$, at a temperature $T = T_c - \Delta T$ below the critical point.



Fig. 4.2. Coarsening of domain structure following quenches from the disordered phase region at $T = 630^{\circ}$ C to the region of two coexisting phases at $T = 570^{\circ}$ C. Two concentrations corresponding to off-critical paths (left and right panels) and a concentration corresponding to a near-critical path (center set of panels) are shown. Reprinted photo 1 with permission Oki *et al.* (1977).

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Reverse Epitaxy of Ge: Ordered and Faceted Surface Patterns

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Normal incidence ion irradiation at elevated temperatures, when amorphization is prevented, induces novel nanoscale patterns of crystalline structures on elemental semiconductors by a reverse epitaxial growth mechanism: on Ge surfaces irradiation at temperatures above the recrystallization temperature of 250 °C leads to self-organized patterns of inverse pyramids. Checkerboard patterns with fourfold symmetry evolve on the Ge (100) surface, whereas on the Ge (111) surface, isotropic patterns with a sixfold symmetry emerge. After high-fluence irradiations, these patterns exhibit well-developed facets. A deterministic nonlinear continuum equation accounting for the effective surface currents due to an Ehrlich-Schwoebel barrier for diffusing vacancies reproduces remarkably well our experimental observations.



Sand Ripples and Dunes

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Does Anticoarsening exist?

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Tuning Ag/Si(100) island size, shape, and density

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Anticoarsening and complex dynamics of step bunches on vicinal surfaces during sublimation

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Important questions

- When does (anti)coarsening occur?
- Why does coarsening may stop?
- What is the coarsening exponent, $L(t) \approx t^n$?

The answers (to the third Q) highlight the importance of

- Conservation laws
- Space dimension and order parameter dimension
- Noise

as shown (for O(n) models) by Bray and Rutenberg...

(Bray&Rutenberg)'s theory of Phase ordering [PRE 49, R27 (1994)]

$$\partial_t \vec{\phi}_k = -k^\mu (\partial \mathcal{H} / \partial \vec{\phi}_{-k}) \qquad \qquad \mathcal{H}[\vec{\phi}] = \int d^d x [(\nabla \vec{\phi})^2 + (\vec{\phi}^2 - 1)^2]$$

"We consider the time dependence of the energy as the system relaxes towards its ground state, using the dynamic scaling hypothesis. We obtain L(t) consistently by comparing the global rate of energy change to the energy dissipation from the local evolution of the order parameter."



Beyond Bray & Ruthenberg theory:

have not a Lyapunov functional some models { cannot be described in terms of defects display interrupted coarsening or anticoarsening are discrete

Our starting point:

Coarsening as phase instability of periodic steady states.

Our approach: a multiscale analysis to get a phase diffusion equation

If $u_0(x)$ is a stationary solution, also $u_0(x + \psi)$ is solution. We study phase dynamics assuming that $\psi = \psi(X, T)$.

Highlights:

- Dynamical infos from static infos
- Systematic derivation of the coarsening exponent, $L(t) \approx t^n$
- Understanding when (and why) coarsening occurs
- Discriminate between perpetual and interrupted coarsening (and anticoarsening?)
- Applicability to non-potential equations



TDGL (mod A) generalized
$$\partial_t u = B(u) + G(u)u_{xx}$$
 ($d = 1, 2$)



CH (mod B) generalized $\partial_t u = -C(u)\partial_{xx}[B(u)+G(u)u_{xx}]$ (d = 1, 2) Crystal growth equation $\partial_t h(\mathbf{x}, t) = -\nabla \cdot [\mathbf{j}(\nabla h) + \nabla(\nabla^2 h)]$ (d = 2) KS conserved $\partial_t u = -\partial_{xx}[u + u_{xx} - (u_x)^2]$



Oono equation $\partial_t u = -\partial_{xx}[B(u) + u_{xx}] - \alpha u$



For all equations, we have obtained

$$\partial_T \psi = \mathbf{D} \partial_{XX} \psi$$

where $D = D(u_{\lambda}(x))$ encapsulates dynamical properties:

•
$$D = \frac{\partial A}{\partial q} < 0 \iff \text{ phase instability}$$

•
$$|D(L)| \approx L^2/t \Rightarrow \text{ coarsening law } L(t)$$

In some cases, we get the following criterion:

coarsening
$$\iff \frac{\partial \lambda}{\partial A} > 0$$

Special dynamical scenarios in 1d (for generalized TDGL and CH)



DNLS

$$H(n,\phi) = \sum_{i} n_i^2 + 2\sum_{i} \sqrt{n_i n_{i+1}} \cos(\phi_i - \phi_{i+1})$$
$$A(n,\phi) = \sum_{i} n_i$$

 $h > 2a^2$ negative temperature region

Simplified (and purely stochastic) DNLS

$$H(n,\phi) = \sum_{i} n_{i}^{2} + 2\sum_{i} \sqrt{n_{i}n_{i+1}} \cos(\phi_{i} - \phi_{i+1})$$

This model produces "breathers" which coarsen endlessly







When does coarsening stop?

•
$$\frac{\partial \lambda}{\partial A} = 0$$
 ($D(q) = 0$)

- Coarsening is driven by a free energy, which has a minimum for a finite domain size: the 1d Ising model or the Oono equation (diblock copolymers).
- Some additional ingredient introduces a new length scale (disorder, stirring/shearing of binary fluids).
 Coarsening may stop or become extremely slow.

Anticoarsening: the first example





Eckhaus instability

 $\partial_t u = \epsilon u - (1 + \partial_{xx})^2 u - u^3$

Another (unacknowledged) example Phase separation in diblock copolymers

$$\partial_t u = -\partial_{xx} [B(u) + u_{xx}] - \alpha u$$



Open, general questions

- Is there coarsening outside a phase instability? (I think not)
- Is there anticoarsening outside a phase instability? (I think yes)
- How to determine *D* for discrete models
- Is it possible to determine *D* from a simulation?
- Is there some principle, based on D, driving the dynamics?

In collaboration with





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Recent results on a growth equation in 2d

$$\partial_t h(\mathbf{x},t) = -\nabla \cdot \left[\mathbf{j}(\nabla h) + \nabla (\nabla^2 h) \right]$$

 $\mathbf{j}(
abla h) =
abla h + ext{nonlinear terms}$

$$\omega(\mathbf{q}) = \mathbf{q}^2 - \mathbf{q}^4$$



Cu(100)/Cu (Zuo & Wendelken, PRL 1995) [1000-5000Å]

Pt(111)/Pt (Th. Michely, Köln University)

Multiscale approach for the growth equation

Perturbative expansion a	and	different scales					
		fast scales	slow scales				
		t = t	$T = \varepsilon^2 t$				
$h = \tilde{h}_0 + \varepsilon \tilde{h}_1 + \dots$		x = x	$X = \varepsilon x$				
		y = y	$Y = \varepsilon y$				
		$arphi_i$	$\psi_i = \varepsilon \varphi_i$				
$\varphi_i = \mathbf{q_i} \cdot \mathbf{x}$	C	$\mathbf{h_i} = \nabla \varphi_i$					
$q_i = q_i(X, T) = \nabla_X \psi_i(X, T)$							
$\partial_t = \varepsilon \left((\partial_T \psi_1) \partial_{\varphi_1} + (\partial_T \psi_2) \partial_{\varphi_1} \right)$	$\varphi_2)$	$\nabla = \nabla$	$7_0 + \varepsilon \nabla_{\mathbf{X}}$				

Phase diffusion equations

$$\begin{cases} \partial_T \psi_1 = \sum_{\alpha \beta \gamma} D^{\mathbf{1}\alpha}_{\beta \gamma} \frac{\partial^2 \psi_\alpha}{\partial X_\beta \partial X_\gamma} \\\\ \partial_T \psi_2 = \sum_{\alpha \beta \gamma} D^{\mathbf{2}\alpha}_{\beta \gamma} \frac{\partial^2 \psi_\alpha}{\partial X_\beta \partial X_\gamma} \end{cases}$$

12 diffusion coefficients $D^{ilpha}_{eta\gamma}$ depend on ${ ilde h}_0$ and its derivatives

For square and 6-fold patterns:

$$\partial_T \psi_1 = D_{11} \frac{\partial^2 \psi_1}{\partial X_1 \partial X_1} + D_{22} \frac{\partial^2 \psi_1}{\partial X_2 \partial X_2} + D_{12} \frac{\partial^2 \psi_2}{\partial X_1 \partial X_2}$$
$$\partial_T \psi_2 = D_{22} \frac{\partial^2 \psi_2}{\partial X_1 \partial X_1} + D_{11} \frac{\partial^2 \psi_2}{\partial X_2 \partial X_2} + D_{12} \frac{\partial^2 \psi_1}{\partial X_1 \partial X_2}$$

Their solutions $\psi_i = \psi_i^0 e^{\Omega T} e^{i\mathbf{K}\cdot\mathbf{X}}$ give the stability, through the dispersion curves: $\Omega_{1,2} = \Omega_{1,2}(\mathbf{K})$

For 6-fold patterns, $D_{11} - D_{22} - D_{12} = 0$

$$\Omega_1(\mathbf{K}) < 0 \qquad \qquad \Omega_2(\mathbf{K}) = -D_{11}K^2$$

$$D_{11} = \frac{2 q^{7/4} \partial_q \left(q^{5/4} \langle h_{11} h_{12} \rangle \right)}{\langle h^2 \rangle}$$

Small amplitude approximation

$$h(\mathbf{x},t) = a_1(t)(e^{iq(x+y)} + e^{iq(x-y)}) + C.C.$$



We can determine the amplitude \tilde{a}_1 for the steady states and D_{11} .

We have evidence of the standard three scenarios:

- Coarsening
- No Coarsening
- Interrupted coarsening

according to the sign of
$$rac{d ilde{a}_1}{dq}$$

Coarsening exponents

There are two universality classes

Faceting (constant slope m)

$$\mathbf{j} = \mathbf{m}(1 - m^2)$$

 $L(t) \approx t^{1/3}$

No faceting (increasing slope m)

$$\mathbf{j} = \frac{\mathbf{m}}{1+m^2}$$

$$L(t) \approx t^{1/4}$$

